# Journal Name

# ARTICLE TYPE

Cite this: DOI: 10.1039/xxxxxxxxx

Received Date Accepted Date

DOI: 10.1039/xxxxxxxxx

www.rsc.org/journalname



Jens Neu,\* Coleen T. Nemes, Kevin P. Regan, Michael R. C. Williams,<sup>‡</sup> and Charles A. Schmuttenmaer\*

DL-norvaline is a molecular crystal at room temperature and it undergoes a phase transition when cooled below 190 K. This phase transition is believed to be Martensitic, thus making it of particular interest for molecular machines. In this paper we investigate this phase transition by measuring its terahertz (THz) spectrum over a range of temperatures. Temperature-dependent THz time-domain spectroscopy (THz-TDS) measurements reveal that the transition temperature ( $T_{\beta \to \alpha}$ ) is 190 K. The influence of nucleation seeds was analyzed by determining the  $T_{\beta \to \alpha}$  of molecular crystals with varying grain size. Grains of  $5\,\mu$ m or less result in a lower transition temperature ( $T_{\beta \to \alpha} = 180$  K) compared to larger grains of  $125\,\mu$ m- $250\,\mu$ m ( $T_{\beta \to \alpha} = 190$  K). Additionally, we gain insight into the physical process of the phase transition via temperature-dependent THz-TDS spectra of doped and mixed molecular crystals. The addition of molecular dopants, which differ from DL-norvaline only at the end of the side chain which resides in the hydrophobic layers of the crystal, decreases  $T_{\beta \to \alpha}$ . This is consistent with a solid-solid phase transition in which the unit cell shifts along this hydrophobic layer, and it leads us to believe that the phase transition in DL-norvaline is Martensitic in nature.

# 1 Introduction and motivation

Most of the proteinogenic amino acids exhibit crystalline forms in the solid state. The same is also true for other non-proteinogenic  $\alpha$ -amino acids such as norvaline (COOHCHNH<sub>2</sub>CH<sub>2</sub>CH<sub>2</sub>CH<sub>3</sub>). In many cases, multiple crystalline solid phases, each with a unique geometry and distinct physical properties, have been identified for a given material. Polymorphism occurs when different crystalline phases (polymorphs) of the same material are thermodynamically accessible.<sup>1-4</sup> Previous studies have observed phase transitions between different polymorphs of various amino acids as a function of temperature or pressure. 5-9 In other amino acids, multiple polymorphs have been found to exist at the same temperature and pressure.<sup>10–12</sup> This is possible because specific crystallization conditions may favor the formation of metastable polymorphs, which are structures representing local, not global, thermodynamic free energy minima at a given temperature and pressure.

Typically, solid-solid phase transitions in molecular crystals (i.e.

E-mail: jens.neu@yale.edu, charles.schmuttenmaer@yale.edu

from one polymorph to another) are understood to take place via a nucleation and growth mechanism. <sup>13–15</sup> In this model, a small nucleus of a new crystalline phase is first formed at a defect site in the original lattice phase. The growth of the new phase proceeds as molecules from the original phase break out of their lattice position and find a place in the new lattice. This rearrangement is linked with a diffusion of molecules in the newly forming lattice. In general, there is no definite relationship between the orientation of the original lattice and the new lattice since the orientation of the new phase's initial nucleus is random or at least highly idiosyncratic.

In rare cases, the transition between two polymorphs is thought to take place via a *concerted*, diffusion-free mechanism, known as a Martensitic or displacive phase transition.<sup>16–19</sup> The resulting process does not cause a randomized reorganization of the lattice, and therefore the lattice exhibits a memory effect.<sup>20</sup> Molecular systems that undergo these concerted phase changes are of particular interest for the understanding and design of molecular machines.<sup>21</sup> Unlike typical solid-solid phase transitions, the concerted motion of a displacive phase transition results in a net force on a microscopic length scale. Therefore, progress towards observing, tuning, or triggering these phase transitions has applications in nanoscale engineering.<sup>22,23</sup>

A standard method of studying solid-solid phase transitions is temperature-dependent X-ray diffraction, which provides essen-



Yale University, Department of Chemistry, PO Box 208107, 225 Prospect St., New Haven, CT 06520-8107, USA.

<sup>†</sup> Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/b000000x/ ‡ Present address: Center for Integrated Nanotechnologies, Los Alamos National Laboratory, Los Alamos, New Mexico 87545, USA.

tial information about a molecular crystal's time-averaged geometrical structure. However, since motion is the characteristic property of displacive phase transitions, it is naturally desirable to complement structural information with dynamical information. For this application, terahertz time-domain spectroscopy (THz-TDS)<sup>24-29</sup>, which allows detection of infrared-active vibrations in the low frequency far-infrared range (typically  $10-200 \,\mathrm{cm}^{-1}$ , and  $1 \text{ THz} = 33.34 \text{ cm}^{-1}$ ) is a promising tool. Most importantly, vibrations in this frequency range can be significantly intermolecular in character<sup>12,30</sup> and are therefore extremely sensitive to the arrangement and orientation of molecules in the crystal lattice. 11,31-33 As a result, THz-TDS measurements can easily distinguish among different polymorphs. Furthermore, an understanding of the specific vibrational motion of the molecules at these low frequencies may provide insight into the mechanism by which a concerted phase change occurs.

# 2 Experimental setup and sample preparation

#### 2.1 Sample Preparation

DL-norvaline was purchased from Alfa Aesar (purity 98%) and used without further purification. The dopant amino acids, which are listed in the Electronic Supplementary Information(Sigma Aldrich, purity 99%), were also used without further purification. 100 mL solutions containing pure DL-norvaline (1.7 g) and 12% per mole of the impurity were prepared. The mixtures slowly evaporated at 40 °C resulting in crystals with sub-millimeter dimensions. The crystallites were characterized with powder XRD at room temperature (Rigaku SmartLab X-ray Diffractometer, see Electronic Supplementary Information). The crystallized materials were ball-milled for 3 min resulting in a homogeneous powder containing grains less than 5 $\mu$ m in size.

The samples used in this study are pressed pellets of a mixture of PTFE (Teflon, purchased from ChemCruz) and a small amount of amino acid (typically  $\approx$ 25 mg amino acid in  $\approx$ 1000 mg Teflon, see Table S1 in the Electronic Supplementary Information). The amino acid crystallites are either pure DL-norvaline , or DL-norvaline crystallized from a mixed solution with the dopant. The mixtures were then pressed with an applied pressure of 11 kbar, and the resulting pellets have a diameter of 13 mm and are approximately  $d \approx 3.3$  mm thick. Reference samples of pure Teflon were pressed under the same conditions.

Teflon-mixed pellets have several advantages over pure amino acid pellets. Thicker samples can be prepared due to the low absorption of Teflon in the THz range. Thicker samples have the advantage of being less fragile and easier to handle, and they also allow one to remove etalon effects in the THz spectrum via truncation of the time-domain measurement. In addition, the fraction of amino acid in the Teflon host material can be optimized such that the absorption features are not saturated.

#### 2.2 Experimental Setup

Spectra were measured in a THz-TDS spectrometer as described in previous publications.<sup>12</sup> For temperature controlled measurements, the samples were mounted in a sample-in-vapor cryostat cooled with nitrogen gas, which allows measurements from room temperature down to 78 K. The cryostat was cooled in either 10 K or 5 K increments using an adjustable nitrogen flow and resistive-heaters. The sample was allowed to reach thermal equilibrium prior to each measurement. To avoid contamination of the sample, a temperature sensor was not placed directly on the sample, but rather above and below it. The average of both temperature sensors provides an accurate approximation of the sample's temperature ( $\pm$  0.5 K).

THz-TDS measures the *amplitude* of the transmitted THz pulse rather than its intensity. Thus, Fourier transformation of the measured time domain traces provides information on both the spectral amplitude and phase of the measured electric field. Given phase and amplitude information of the THz pulse, the real (n')and imaginary (n'') parts of the complex-valued refractive index can be obtained directly.

#### 2.2.1 Extraction of the absorption-coefficient from the measured data

The complex transmission amplitude *t* of the sample is described by  $^{34}$ :

$$t(\boldsymbol{\omega}) = t_{12}(\boldsymbol{\omega}) \cdot p_2(\boldsymbol{\omega}) \cdot t_{21}(\boldsymbol{\omega})$$
(1)

where  $t_{ij}(\omega) = \frac{2n_i(\omega)}{n_i(\omega)+n_j(\omega)}$  is the Fresnel amplitude transmission coefficient from medium *i* to medium *j*,  $p_2 = e^{ik_0dn_2}$  is the propagation operator through the sample (medium 2) where  $k_0 = (\omega/c)$ is the wave-vector in vacuum, and *d* the sample thickness. In the present case, the THz pulse enters the sample from air  $(n_1 = 1)$ , propagates through the sample, and then exits at the sample-air interface. The pellets are d > 3 mm thick, which corresponds to more than 30 ps for the doubly-reflected pulse to emerge. Therefore, a measurement time duration of 28 ps was chosen in order to avoid multiple pulses, and thus eliminate etalon effects in the complex transmission function.

The measured transmission through the sample is complexvalued and can therefore be separated into an amplitude and phase term. This allows us to rewrite eq. (1) as:

$$t(\omega) = t_{12}(\omega)t_{21}(\omega)e^{-k_0dn_2''}e^{+idk_0n_2'} = Ae^{-i\phi \pm 2m\pi}$$
(2)

with the complex refractive index n = n' + in''. It is possible to calculate the real part of the refractive index based on the measured spectral phase.<sup>35</sup> From this refractive index it is possible to calculate the transmission coefficient and then solve eq. (2) for the imaginary part of the refractive index.

Effective medium theory is used to calculate the index for the amino acid. The well-known Bruggeman equation<sup>36</sup> relates the measured permittivity of a mixture ( $\varepsilon_m$ ) to the permittivity of the two components ( $\varepsilon_h$  for Teflon host, and  $\varepsilon_i$  for amino acid inclusions):

$$f_h\left(\frac{\varepsilon_h - \varepsilon_m}{\varepsilon_h + 2\varepsilon_m}\right) + f_i\left(\frac{\varepsilon_i - \varepsilon_m}{\varepsilon_i + 2\varepsilon_m}\right) = 0 \tag{3}$$

where  $f_h$ ,  $f_i$  are the volume fraction of the two components, h (host), and i (inclusion). The measured refractive index ( $n_m = n_2$ ) describes the mixed pellet. The permittivity of the host material  $\varepsilon_h = n_{\text{Teflon}}^2$  is calculated from the reference measurement, and the permittivity of the mixture  $\varepsilon_m = n_m^2$  is calculated from the mixed

pellet, under the assumption that the magnetic response of the medium is negligible (relative permeability  $\mu_r = 1$ ). Therefore, eq. (3) can be solved for the complex permittivity of the amino acid ( $\varepsilon_i$ ). The complex refractive index of the amino acid crystallites is given by  $n = \sqrt{\varepsilon_i}$ , and the absorption coefficient is determined from the imaginary part:  $\alpha = 2n'' \frac{\omega}{c}$ .

#### 2.3 DFT Calculations of DL-norvaline

Calculations of the optimized structure and vibrational modes of DL-norvaline were carried out using the SIESTA DFT software package (3.2 revision 462). We employed the vdW-DF-cx functional<sup>37</sup>, which is based on a generalized gradient approximation (GGA) level exchange-correlation functional, but which also accounts for van der Waals forces using the Lundqvist-Langreth nonlocal density-density interaction framework.<sup>38</sup> Typical GGA or meta-GGA exchange-correlation functionals are capable of modeling only local or semi-local forces, and this is inadequate for modeling a system with regions dominated by hydrophobic interactions, and therefore van der Waals (dispersion) forces, such as DL-norvaline. These DFT calculations incorporate vdW-DF functionals that do not require empirical correction terms. In particular, the vDW-DF-cx functional aims to be nonempirical and to use the vdW-DF scheme to calculate both correlation and exchange terms in a self-consistent way.<sup>37</sup>

SIESTA uses periodic boundary conditions, and the basis set consists of numerical atomic orbitals. We used a triple-zeta, doubly polarized (TZDP) basis set. Minimum energy structures were obtained by optimizing the initial coordinates and lattice cell parameters obtained by XRD measurements on single crystals.<sup>39</sup> The atomic positions and unit cell parameters were simultaneously optimized to minimize interatomic forces and unit cell stress. Details of calculating vibrational frequencies and IR intensities has been previously reported. <sup>12</sup> Briefly, harmonic modes were calculated using the force-constant matrix<sup>40</sup> determined by displacing each atom 0.02 Å in each direction along the Cartesian axes. For IR intensities, the Berry phase approach was used to calculate macroscopic polarization<sup>41</sup>, which was then used to calculate the Born effective charge tensor of each atom. 40 The Born effective charge tensor of each atom was then used to calculate the IR intensity of each mode. 12,41

# **3** Experimental Results

#### 3.1 Pure DL-norvaline

A Teflon/DL-norvaline pellet was cooled in 10 K increments. At selected temperatures, the transmission though the sample pellet was measured along with the transmission through an identical aperture without a sample (the reference). The absorption spectra, calculated from these measurements, are plotted in Fig. 1. For temperatures higher than 180 K, the samples show a broad, weak absorption with a central-frequency of 1.65 THz and full width at half maximum (FWHM) of about 0.8 THz. The THz transmission spectrum changes fundamentally as the sample is cooled from 190 K to 180 K. A strong and narrow peak emerges at 1.55 THz with a FWHM of 0.26 THz. THz measurements are sensitive to collective vibrational and rotational/librational modes in



Fig. 1 Measured absorption coefficient of DL-norvaline. The absorption spectrum changes dramatically between 190 K and 180 K, indicating that the crystal lattice changes at this temperature. The absorption increases with decreasing temperature, and reaches a maximum at 140 K. The absorption then decreases as the temperature is further decreased.

the crystal lattice. In general, these modes are narrower and more pronounced at lower temperatures.<sup>31</sup> The distinct change in the absorption mode seen here, and in particular the red-shift of the peak with decreasing temperature, cannot be explained merely by thermal shrinkage of the lattice. Such a significant change in the transmission spectrum can only be explained by a similarly significant change of the lattice, which is a solid-solid phase transition that occurs between 190 K and 180 K and has also been observed in single crystal XRD measurements.<sup>39</sup>

The absorption spectrum continues to change as the temperature is decreased from 180 K to 140 K. The absorption strength increases, and the characteristic feature becomes narrower. For temperatures below 140 K, however, the absorption decreases with decreasing temperature, and this also cannot be explained by a simple contraction of the lattice. Although the change in the absorption spectrum is more subtle than that between 190 K and 180 K, it indicates a second solid-solid phase transition. The phase transition between 190 K and 180 K is reversible, but the one starting at around 140 K is irreversible. This is in agreement with delamination in DL-norvaline observed at these lower temperatures.<sup>39</sup>

#### 3.2 Comparison of Calculated and Measured THz Spectra

The frequencies and IR intensities at 0 K were calculated after optimizing the atomic coordinates determined crystallographically at -90 °C (183 K).<sup>39</sup> In accordance with previous studies<sup>42</sup> the D and L enantiomers alternate along the *c* axis, and are identical along the *a* and *b* axes.<sup>39</sup> For the low temperature  $\alpha$  phase, norvaline molecules adopt two conformations, denoted A and B, where A had an abundance of 51.5%.<sup>39</sup> The side chain can adopt trans and gauche conformations upon rotation around the C<sup> $\alpha$ </sup>-C<sup> $\beta$ </sup> and C<sup> $\beta$ </sup>-C<sup> $\gamma$ </sup> bonds. The A conformer of the  $\alpha$  phase is trans,gauche+,



**Fig. 2** Optimized DL-norvaline unit cell with conformer configuration AABB. Molecules in positions 1 and 3 are L, and those in positions 2 and 4 are D.

and the B conformer is trans,trans.<sup>39</sup> The reported single crystal x-ray diffraction data (CCSD zzzodu02) does not determine which molecule in the unit cell is the A conformer and which is B.<sup>39</sup>

Fig. 2 shows the optimized geometry of a 4-molecule unit cell where A-B neighbors are along the a axis, A-A and B-B neighbors are along the c axis, and A-A and B-B neighbors are along the baxis. We denote each type of unit cell by the sequence of conformer type in positions 1-2-3-4. For example, Fig. 2 displays the AABB unit cell. We note that the structural optimization does not result in conformational changes of the molecules. Unit cell calculations included all six possible conformer configurations with a 50% abundance of each: AABB, BBAA, ABBA, BAAB, ABAB, and BABA. The results of these calculation are presented in table 1.

The initially monoclinic symmetry of the unit cell is broken during the optimization of the ABBA and BAAB configurations, and their minimum energy calculated structures are triclinic. Furthermore, these configuration deviate strongly from the previously reported geometry. The *a* axis is significantly longer in the calculation than in previously reported crystallographic data. Therefore, we conclude that these two conformer combinations are less likely. The remaining four unit cells (ABAB, BABA, AABB, BBAA) all optimized to axes length within  $\pm 2\%$  of the reported crystallographic dimensions. This slight discrepancy between the calculation and previous crystallographic results is to be expected, because the crystallographic structure is determined at 183 K, while the calculation is at 0 K. The final unit cell energy of the ABAB conformer is the lowest, hence the most favorable. However, the energetic difference of 0.13 eV (3.0 kcal/mol) is not large enough to rule out the remaining three unit cells.

The IR spectrum for all six unit cells is calculated and plotted in the Electronic Supplementary Information (Fig. S1 and S2). The calculated spectra all exhibit a resonance around 1.5 THz, in agreement with our experimental results. To better understand the effect of conformer configuration on the calculated spectrum, we performed a mode character analysis, in which intermolecular and intramolecular contributions of each mode are quantified.<sup>31</sup> As seen in the SI Fig.S3, modes across different conformer configurations can be grouped according to similar frequency, intensity, and mode character. The intense peaks in the 1.4 THz to 1.75 THz range are mostly rotational/librational in character (blue and green squares), while those in the 2.0 THz to 2.4 THz range have equal contributions from rotational/librational and intramolecular motions with almost no translational character (yellow squares). In general, greater similarities are observed for conformer configurations with similar optimized unit cell parameters, for example AABB and BBAA.

The spectrum calculated for the ABAB conformer combination, is plotted in Fig. 3. The calculated resonances are convoluted with a Lorentzian function with FWHM of 0.2 THz to mimic the experimental line width. The strong peak is very well reproduced by the calculations. The broad minor peak close to 2 THz is also reasonably well reproduced.

#### 3.3 Grain Size Dependence

A more detailed investigation of the phase change between 190 K and 180 K can be performed by measuring the phase transition of DL-norvaline / Teflon pellets with differing DL-norvaline grain sizes. If we assume that the defect and impurity density is independent of grain size, smaller grains will contain fewer defects and impurities. These impurities and defects act as nucleation points for phase transitions. This holds true for both the nucleation and growth as well as Martensitic mechanisms of phase transitions.  $^{17,43,44}$  A larger total number of impurities/defects will disturb the ideal lattice to a greater extent, and the energy barrier for a phase transition will be lower.

We measured THz spectra of samples with two different grain

	1.04			1551		1212	
Geometry / Label	zzzodu01	AABB	BBAA	ABBA	BAAB	ABAB	BABA
<i>a</i> [Å]	14.21	14.27	14.27	14.54	14.53	13.91	13.91
<i>b</i> [Å]	4.76	4.86	4.86	4.88	4.88	4.87	4.87
<i>c</i> [Å]	9.86	9.84	9.84	9.80	9.80	8.82	9.82
α [°]	90	90	90	89.66	90.35	90	90
β [°]	102.85	102.93	102.95	102.83	102.83	102.22	102.17
γ [°]	90	90	90	92.55	87.53	90	90
Volume [Å <sup>3</sup> ]	649.81	665	665.41	677.19	676.90	650.42	650.41
Energy [eV]		-8360.06	-8360.06	-8360.03	-8360.03	-8360.16	-8360.16

Table 1 Geometry of the unit cells calculations performed with SIESTA. The labels denote the position of the conformer in the unit cell, based on the notation defined in Fig 2.



Fig. 3 Calculated IR intensity of the ABAB conformer combination. The calculated IR resonances are broadened with a Lorentzian with FWHM of 0.2 THz. The blue dashed line is the experimentally measured spectrum at 140 K.

sizes to investigate this. The first sample contained crystals that were ball-milled for 3 min to produce grains smaller than  $s = 5 \mu m$  in diameter. For the second sample, the grains where ground using a mortar and pestle and sorted by size using a micro sieve. Grain diameters of this second sample ranged from  $125 \mu m$  to  $250 \mu m$ .



**Fig. 4** Measured absorption through a sample made of pure DL-norvalinewith an average grain size *s* of about  $5 \,\mu$ m. A clear change in the spectral absorption is evident when the sample is cooled from 185 K to 180 K.

The results for the ball-milled sample are plotted in Fig. 4. A temperature step size of  $5 \text{ K} \pm 0.5 \text{ K}$  was chosen to more accurately determine the phase transition temperature. When the sample was cooled from 185 K to 180 K, the absorption profile changed abruptly. Measurements performed on the larger grains are plot-



**Fig. 5** Measured absorption through a sample made of pure DL-norvaline, with an average grain size *s* between  $125 \,\mu\text{m}$  and  $250 \,\mu\text{m}$ . A clear change in the spectral absorption is evident when the sample is cooled from  $195 \,\text{K}$  to  $190 \,\text{K}$ .

ted in Fig. 5. A clear change in the spectrum is visible when the sample is cooled from 195 K to 190 K, which is 10 K higher than for the micro-grain sample in Fig. 4. The only difference in these samples is their grain size, and therefore the total number of imperfections and defects per grain. The larger grains have an about 50,000 times larger volume, and under the assumption of a constant imperfection density, the probability of having an imperfection is 50,000 times higher. The resulting 10 K difference in phase transition temperature suggests that these imperfections are important as starting points for the phase transition, which is in agreement with a Martinsitic phase transition mechanism, but does not rule out nucleation and growth.

A comparison of the two different grain sizes also reveals that the high temperature absorption coefficient of both samples is  $\alpha = 145 \pm 15 \text{ cm}^{-1}$ , however the low temperature absorption at 175 K is different. It is  $\alpha = 210 \pm 15 \text{ cm}^{-1}$  for the small grain size, while it is  $\alpha = 310 \pm 15 \text{ cm}^{-1}$  for the larger ones. This indicates that possibly not all grains in the micro granular sample have converted to the low temperature phase, even through 175 K is more than 15 K colder than the phase transition temperature measured for the larger grains.

#### 3.4 Mixed amino acid crystals

A more detailed understanding of the origin of the resonant absorption and the phase transition can be gained by measuring doped DL-norvaline crystals. The molecular dopants have similar geometries and chemical properties as the host material. A concentration of 12 mol-% dopants in solution was chosen. This high concentration ensures that enough material for the crystallization process is present to form not only doped crystals but also mixed crystals.<sup>45</sup> In general, three different types of crystal can result from this mixing. The first case is that the two amino acids form two pure crystals that are commingled. The PXRD of this crystal



**Fig. 6** PXRD of DL-norvaline and DL-norvaline doped with L-glutamic acid. The blue and the green lines correspond to the two pure materials, and the black line is the measured spectrum of the doped material. The clear shift between the experimental spectral features and the features of the pure DL-norvaline, correspond to an increased lattice constant of the mixed crystal.

would be a simple linear combination of the pure amino acids. This case has not be detected. The second case would be a formation of a doped crystal, that is, where 12% of the original lattice molecules are replaced by the dopant. The third case would be the formation of a different type of mixed crystal that has regions of pure DL-norvaline crystals and other regions of crystals formed by 50:50 or 75:25 mixtures in the unit cell.

Powder X-ray diffraction (PXRD) measurements were performed to determine whether the dopant is incorporated in the host lattice or forms a mixed crystal. The PXRD spectra of DL-norvaline, L-glutamic acid, and DL-norvaline doped with Lglutamic acid, are plotted in Fig. 6. The intensity of the peaks for the pure crystals are weighted with their molar concentration. If the crystal consists of regions of pure DL-norvaline and regions of pure dopant, the corresponding PXRD spectrum would be a simple linear combination of these individual spectra. A small shift of the features would correspond to the doped case<sup>46</sup>, while new features, existing in neither the dopant nor in DL-norvaline, can be explained by the formation of a new lattice. This new lattice case is referred to as mixed crystals.

The PXRD spectrum of the crystal in Fig. 6 shows similar features as the pure material. These features are shifted to slightly smaller angles and therefore we conclude that the lattice is slightly larger than the lattice of the pure material. This result is consistent with a doped crystal in which the dopant material occupies some of the lattice positions of the pure materials by replacing them.<sup>46</sup> This behavior was observed for DLnorvaline doped with DL-norleucine, L-glutamic acid, and Lphenylglycine (see Electronic Supplementary Information, Figure S5). On the other hand, the PXRD spectra for DL-norvaline mixed with equal amounts of L-phenylglycine and D-phenylglycine, and L-asparagine have additional features that are neither in the pure



Fig. 7 Measured absorption through a sample made of DL-norvaline doped with L-glutamic acid. The absorption characteristics of the sample change between 180 K and 170 K. However, compared to the pure DL-norvaline, this change is more gradual, which suggests that not all grains in the sample undergo a phase transition at the same time.

dopant spectra nor in the pure DL-norvaline spectra (see Electronic Supplementary Information, Figure S6). These new features are explained by a fraction of the sample is a mixture that forms a new lattice which is different from the lattice of the pure DL-norvaline. Therefore, these crystals are referred to as mixed crystals to distinguish them from the doped crystals.

#### 3.4.1 DL-norvaline doped with L-glutamic acid

The measured absorption spectrum of DL-norvaline doped with L-glutamic acid is plotted in Fig. 7. The overall absorption at temperatures above the phase transition is comparable to that for pure DL-norvaline. There is a broad absorption feature around 1.6 THz with a FWHM of 0.7 THz. A phase change occurs when the sample is cooled to a temperature of less than 170 K. The broad resonance changes into a narrow resonance located at 1.5 THz whose size and spectral width are comparable to the measured resonance of the pure material. The strength of this resonance increases with decreasing temperature, indicating that more grains are transforming into the low temperature phase. This is in strong contrast to the pure material, in which a second phase transition reduces the absorption for temperatures below 140 K.

In combination with the PXRD results, we conclude that the dopant molecules are incorporated in the original lattice and do not significantly change the resonance frequency. However, the dopant influences the phase transition temperature, which is reduced to 170 K. The dopant provides additional lattice defects, acting as seeds for a crystallization in the nucleation and growth mechanism, and therefore, the phase transition temperature is expected to be higher. The fact that the phase transition occurs at a lower temperature is consistent with it being Martensitic. In this transition, the unit cell slides along the *b* axis, by half a unit cell length.<sup>39</sup> This sliding is a collective, quasi-simultaneous movement of all molecules in the layer. The -R group in L-glutamic



Fig. 8 Measured absorption through a sample made of DL-norvaline doped with L-phenylglycine. The absorption characteristics of the sample change between 180 K and 170 K. The absorption characteristic of this sample is similar to the pure DL-norvaline, however the maximum absorption is weaker.

acid terminates with a carboxylic acid, but is a methyl group in DL-norvaline. This carboxylic acid group is located in the plane at which the collective displacement takes place, and will locally change the energy profile and thereby hinder this sliding process. The larger potential barriers due to the dopants result in a lower onset temperature.

### 3.4.2 DL-norvaline doped with L-phenylglycine

The measured absorption spectrum for DL-norvaline doped with L-phenylglycine is plotted in Fig. 8, and is similar to the undoped DL-norvaline. The resonance frequency of the low temperature phase is 1.5 THz, which is identical to the resonance frequency of the pure DL-norvaline. However, the phase transition temperature in the doped sample sets in at 170 K rather than 180 K. As the sample is further cooled to 90 K the increase in absorption is less than that for pure DL-norvalineor DL-norvaline doped with L-glutamic acid. Similar to the sample doped with L-glutamic acid, we conclude that the dopants do not completely change the lattice but stabilize it with respect to the second low temperature phase change.

As discussed with the L-glutamic acid doped case, the molecular structure for L-phenylglycine is similar to DL-norvaline, except for the phenyl ring at the end of the -R group which is located at the sliding interface. As in the case for L-glutamic acid, the additional disturbance along this plane, increases the energy barrier and decreases the phase transition temperature.

## 4 Conclusions

DL-norvaline forms a layered crystal and is suspected to undergo a Martensitic phase transition when cooled.<sup>39</sup> This phase transition has been studied with temperature-dependent THz spectroscopy. When the pure DL-norvaline is cooled below 180 K a significant change in the absorption spectrum is measured which is associated with a phase-transition. A better understanding of the mechanism for this phase transition is gained by measuring different crystal sizes since larger crystals have a higher probability for containing impurities. These impurities act as seeds for phase transitions, and therefore larger crystals undergo the phase transition at higher temperatures. For grains with an average size between 125 and  $250\,\mu\text{m}$ , the transition temperature was 190 K, while it is 180 K for grains of  $5\,\mu\text{m}$  or less. This effect is consistent with Martensitic transistions.

The phase transition mechanism was studied in more detail by doping DL-norvaline crystals with similar amino acids. These dopants *reduced* the phase transition temperature by 10 K, although one would expect the additional defect sites caused by the dopants should increase the transition temperature. This is a strong indicator for a Martensitic phase transition, because the suspected transition can be described as a sliding of neighboring unit cells. The dopants used are identical to DL-norvaline, except for the end of the -R group which is found at the hydrophobic layer where the sliding occurs. The dopants locally increase the energy barrier for the sliding process which requires a lower temperature for the Martensitic transition.

## Acknowledgement

We acknowledge financial support from the national science foundation (NSF) under grant number: NSF CHE - CSDMA 1465085. KR and CN acknowledge financial support by the U.S. Department of Energy Office of Science, Office of Basic Energy Sciences, under award no. DE-FG02-07ER15909.

## References

- W. C. McCrone, in <u>Physics and Chemistry of the Organic Solid</u> <u>State. Volume 2</u>, ed. D. Fox, M. M. Labes and A. Weissberger, Interscience Publishers, 1965, pp. 725–767.
- 2 J. Bernstein, <u>Polymorphism in Molecular Crystals</u>, Oxford University Press, New York, 2002, vol. 14.
- 3 A. Gavezzotti, J. Pharm. Sci., 2007, 96, 2232-2241.
- 4 A. D. Bond, <u>Curr. Opin. Solid State Mater. Sci.</u>, 2009, **13**, 91– 97.
- 5 E. V. Boldyreva, H. Sowa, Y. V. Seryotkin, T. N. Drebushchak, H. Ahsbahs, V. Chernyshev and V. Dmitriev, <u>Chem. Phys. Lett.</u>, 2006, **429**, 474–478.
- 6 A. S. Sabino, G. P. De Sousa, C. Luz-Lima, P. T. C. Freire, F. E. A. Melo and J. M. Filho, <u>Solid State Commun.</u>, 2009, 149, 1553–1556.
- 7 V. S. Minkov, A. S. Krylov, E. V. Boldyreva, S. V. Goryainov, S. N. Bizyaev and A. N. Vtyurin, <u>J. Phys. Chem. B</u>, 2008, **112**, 8851–8854.
- 8 V. S. Minkov, N. A. Tumanov, R. Q. Cabrera and E. V. Boldyreva, <u>CrystEngComm</u>, 2010, **12**, 2551–2560.
- 9 E. A. Belo, J. A. Lima Jr., P. T. C. Freire, F. E. A. Melo, J. M. Filho, H. N. Bordallo and A. Polian, <u>Vib. Spectrosc.</u>, 2010, 54, 107–111.
- 10 H. N. Bordallo, E. V. Boldyreva, A. Buchsteiner, M. M. Koza and S. Landsgesell, J. Phys. Chem. B, 2008, 112, 8748–8759.
- 11 A. B. True, K. Schroeck, T. A. French and C. A. Schmutten-

maer, J. Infrared, Millimeter, Terahertz Waves, 2011, **32**, 691–698.

- 12 M. R. C. Williams, D. J. Aschaffenburg, B. K. Ofori-Okai and C. A. Schmuttenmaer, <u>J. Phys. Chem. B</u>, 2013, **117**, 10444– 10461.
- 13 Y. V. Mnyukh, Mol. Cryst. Liq. Cryst., 1979, 52, 505-521.
- 14 J. D. Dunitz, Pure Appl. Chem., 1991, 63, 177–185.
- 15 F. H. Herbstein, Acta Crystallogr., Sect. B, 2006, 62, 341–383.
- 16 S. C. Tuble, J. Anwar and J. D. Gale, <u>J. Am. Chem. Soc.</u>, 2004, 126, 396–405.
- 17 J. Anwar, S. C. Tuble and J. Kendrick, <u>J. Am. Chem. Soc.</u>, 2007, **129**, 2542–2547.
- 18 D. Zahn and J. Anwar, RSC Adv., 2013, 3, 12810.
- 19 J. A. van den Ende and H. M. Cuppen, <u>Cryst. Growth Des.</u>, 2014, **14**, 3343–3351.
- 20 J. Mohd Jani, M. Leary, A. Subic and M. A. Gibson, <u>Materials</u> & Design, 2014, 56, 1078–1113.
- 21 K. Bhattacharya and R. D. James, Science, 2005, 307, 53–54.
- 22 J. S. Juan, M. L. No and C. A. Schuh, <u>Nat Nano</u>, 2009, 4, 415–419.
- 23 K. Kinbara and T. Aida, Chem. Rev., 2005, 105, 1377–1400.
- 24 M. van Exter and D. Grischkowsky, <u>Applied Physics Letters</u>, 1990, **56**, 1694–1696.
- 25 J. Neu, R. Beigang and M. Rahm, <u>Applied Physic Letters</u>, 2013, **103**, 041109–4.
- 26 J. A. Zeitler, P. F. Taday, D. A. Newnham, M. Pepper, K. C. Gordon and T. Rades, <u>Journal of Pharmacy and Pharmacology</u>, 2007, **59**, 209–223.
- 27 J. Neu and M. Rahm, Optics Express, 2015, 23, 12900–12909.
- 28 D. Grischkowsky, S. Keiding, M. van Exter and C. Fattinger, <u>J.</u> Opt. Soc. Am. B, 1990, 7, 2006–2015.
- 29 J. Neu, B. Krolla, O. Paul, B. Reinhard, R. Beigang and M. Rahm, <u>Optics Express</u>, 2010, **18**, 27748–27757.
- 30 P. U. Jepsen and S. J. Clark, <u>Chem. Phys. Lett.</u>, 2007, **442**, 275–280.

- 31 M. R. C. Williams, A. B. True, A. F. Izmaylov, T. A. French, K. Schroeck and C. A. Schmuttenmaer, <u>Phys. Chem. Chem.</u> <u>Phys.</u>, 2011, **13**, 11719–11730.
- 32 M. D. King, W. D. Buchanan and T. M. Korter, <u>J. Pharm. Sci.</u>, 2011, **100**, 1116–1129.
- 33 A. M. Fedor, D. G. Allis and T. M. Korter, <u>Vibrational</u> <u>Spectroscopy</u>, 2009, **49**, 124 – 132.
- 34 S. J. Orfanidis, <u>Electromagnetic Waves and Antennas</u>, Rutgers University, 2006.
- 35 This assumes that there is no phase shift at the interfaces. This is a valid assumption because this additional phase is roughly 1000 times smaller than that from propagation through the sample, as explained in detail in the ESI of this paper.
- 36 T. Choy, <u>Effective Medium Theory: Principles and</u> Applications, Clarendon Press, 1999.
- 37 K. Berland and P. Hyldgaard, Phys. Rev. B, 2014, 89, 035412.
- 38 M. Dion, H. Rydberg, E. Schröder, D. C. Langreth and B. I. Lundqvist, Phys. Rev. Lett., 2004, **92**, 246401.
- 39 C. H. Görbitz, J. Phys. Chem. B, 2011, 115, 2447-2453.
- 40 D. Fernández-Torre, R. Escribano, T. Archer, J. M. Pruneda and E. Artacho, <u>J. Phys. Chem. A</u>, 2004, **108**, 10535–10541.
- 41 R. D. King-Smith and D. Vanderbilt, <u>Phys. Rev. B</u>, 1993, 47, 1651–1654.
- 42 P. Chatzigeorgiou, N. Papakonstantopoulos, N. Tagaroulia, E. Pollatos, P. Xynogalas and K. Viras, J. Phys. Chem. B, 2010, 114, 1294–1300.
- 43 W. Cao, J. A. Krumhansl and R. J. Gooding, <u>Phys. Rev. B</u>, 1990, **41**, 11319–11327.
- 44 A. A. Boulbitch and P. Tolédano, Phys. Rev. Lett., 1998, 81, 838-841.
- 45 J. C. Givand, A. S. Teja and R. W. Rousseau, <u>AIChE J.</u>, 2001, 47, 2705–2712.
- 46 H. C. Koolman and R. W. Rousseau, <u>AIChE J.</u>, 1996, **42**, 147– 153.